

ELABORATION AND CHARACTERIZATION OF A COMPLEX COATING ON Ti WITH TiO₂ NANOTUBES, FUNCTIONALIZED SINGLE CARBON NANOTUBES, HYDROXYAPATITE AND IRON

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Abstract. *The aim of this work is elaboration and characterization of a complex coating on Ti with TiO₂ nanotubes, functionalized single carbon nanotubes, hydroxyapatite (HA) and iron (Fe). The single carbon nanotubes (SWCNTs) were functionalized with carboxyl groups (SWCNT-COOH) and TiO₂ nanotubes were elaborated by anodization. The complex hybrid material with iron immobilized was obtained by chronopotentiometric method. The electrodeposited HA/ SWCNT-COOH/Fe coatings were investigated for two different concentration of Fe as 0.01 and 0.025 M respectively. The coating structures were studied by X-Ray Diffraction. Surface analysis was completed with contact angles determination in order to establish the hydrophilic/hydrophobic balance. Electrochemical determinations were followed by potentiodynamic measurements conducted in Hank solution and by electrochemical impedance spectroscopy investigations(EIS). As biological hemolysis experimentes were performed, and no sign of evident toxic efect was observed.*

Keywords: TiO₂ nanotubes, SWCNTs, hydroxyapatite, electrochemical tests, hemolysis.

1. Introduction

One of the most important topic of 21 century is the bioimplant design in order to obtain materials with remarkable properties. Those can be used in bone and joint replacements, fixation devices or dental implants to make our life easier.

There are many types of materials used to manufacture implantable devices, each one of them having special purposes depending on their composition. They can be

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metallic biomaterials, polymers or ceramics materials. The main merit of these materials is high biocompatibility (non-toxic, non-allergic) with human tissue [2].

Magnesium, iron and zinc base materials have been studied for this purpose. Iron is an essential element for human body, and is used also for the transfer of oxygen by blood [3] and have many others applications, like catalysts for growing carbon nanotubes or photodecomposing organics [4].

Compared to magnesium porous structure, iron posses higher strenght which leads to a better control over the porous structure[5], and also have a better ductility and radio-opaqueness, than magnesium and it's alloys [6], but iron presents some problems due to it's ferromagnetic behaviour and slow degradation rate [7].

Iron oxide is used used as a consequence of it's superparamagnetic properties as contrast agent for diagnostic imaging and magnetic drug delivery [8,9].

Titanium is among the most used scaffold for bioimplant applications as a result of his good corrosion resistance and high strength to be used as load-bearing. An inexpensive method to improve Ti bioactivity represents the formation of Titanium oxide nanotubes via anodization method [10].

A simple way to improve the osseointegration is to cover the metallic implants (Ti) with Hydroxyapatite [11]. This is the major mineral component of human hard tissue, having the ratio of Ca/P of 1.67 which helps to enhance bone recovery and is distinguished by his excellent biocompatibility and low price [12,13].

The researchers tried to incorporate hydroxyapatite with different compounds e.g.: iron ions, carbon nanotubes, etc. in order to obtain materials with better mechanical properties. The iron ions incorporation into hydroxyapatite can be obtained via sol-gel method [14], wet precipitation method [15], or by using the wet chemical method [16]. Gamal et al. demonstrated that the presence of iron addtives in HA structure is leading to the appearance of the point defects obtaining a new hard, strong and durable material.

K.M.T. Ereiba et al. developed FeHAp samples through wet chemical methods and established that iron improved the bio- activity and solubility of HAp under the physiological conditions [17].

The surfaces changes and immobilization of different biomolecules [18] [19], ions [20], nanoparticles [21], nanotubes [22] on biomaterials surface has been applied to improve the biocompatibility of implants. The biological response of biomaterial has a close relationship with the interaction between cells and surfaces.

Platelet adhesion and endothelialization rates are frequently used to assess the biocompatibility of biomaterials [23]. Platelets are disc shaped blood cells emitting pseudopodia when activated in certain conditions .

Red blood cells can be affected also, the interaction between erythrocytes and the materials may lead to certain changes in cell surface features, even more some materials have hemolytic action on normal human erythrocytes in vitro.

Many attempts appear in literature regarding CNTs reinforced Hydroxyapatite as CNTs have caught the attention in medical field due to their unique structure, high surface area, electrical conductivity and low weight [24]. These nanocomposites can be used for bone replacements and could be developed via ion exchange reaction or sol-gel process [25] .

The present paper aims to obtain by chronopotentiometric method a coating material based on TiO₂ nanotubes, hydroxyapatite, iron ions and functionalized single wall carbon nanotubes with carboxyl groups (SWCNT-COOH). We proposed to establish the SWCNTs-COOH influence from electrochemical study in Hank solution and also their biological performance. In our previous work [26] 2 materials containing TiO₂ nanotubes, hydroxyapatite and iron ions with two different concentration (0.01M and 0.025M) were developed. Analyses of the iron behaviour via electrochemical study in Hank solution showed that the increase of iron concentration lead to a higher corrosion rate. We compared the biological behaviour of the new materials with our 2 samples previously developed (TiO₂-HA-Fe 0.01 , TiO₂-HA-Fe 0.025).

2. Materials and methods

2.1. Functionalization of SWCNTs with –COOH groups

SWCNTs functionalization [27] is necessary in order to introduce -COOH groups . SWCNTs are from Sigma Aldrich and have diameter between 20-40 nm and length from to 10 μ m.

The functionalization was performed using aqua regia 2:3 (V:V), 8 h at 50 degree. After this step the samples were washed with ultrapure water and dried.

2.2 Anodization of Ti plates

The titanium sample (99.7% purity, 2mm thick, Sigma-Aldrich) has a selected area of 0.385 cm² . The electrochemical anodizing of titanium samples was performed in order to obtain TiO₂ nanoarchitectures . We used an electrochemical cell with 2 electrodes, with Platinum sheet as a counter electrode, the voltage was kept constant at 20 V, for 1.5 h at room temperature. Anodizing was carried out using HF 0.5% and Na₂HPO₄ 5g/L electrolyte. [26].

2.3. Electrodeposition of HA/ SWCNT-COOH/Fe hybrid material

The electrochemical cell was consisted from anodic titanium nanotubes obtained above, a platinum counter-electrode and with Ag/AgCl as a reference by chronopotentiometric method for 30 minutes. The electrolyte had the following composition: 2 g/L SWCNT-COOH, 9.91 g/L $\text{Ca}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$, and 2.875 g/L $\text{NH}_4\text{H}_2\text{PO}_4$. We used two different concentration of $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ 0.01M and 0.025 M, which was added into the solution mentioned above.

We will refer to 4 samples: sample 1 - $\text{TiO}_2\text{HA-Fe}$ 0.01M electrodeposition of HA-Fe 0.01M on TiO_2 nanotubes obtained by anodization on titanium, sample 2 - $\text{TiO}_2\text{HA-Fe}$ 0.025M electrodeposition of HA-Fe 0.025M on TiO_2 nanotubes obtained by anodization on titanium, sample 3 - $\text{TiO}_2\text{HA-Fe-Swcnt}$ 0.01M electrodeposition of HA- Fe 0.01M and SWCNTs-COOH on TiO_2 nanotubes obtained by anodization on titanium, sample 4 - $\text{TiO}_2\text{HA-Fe-Swcnt}$ 0.025 M electrodeposition of HA-Fe 0.025M and SWCNTs-COOH on TiO_2 nanotubes obtained by anodization on titanium.

2.4. Surface characterization

In order to examine the formation of HA layer also the presence of iron and SWCNTs-COOH we used the bellow methods:

2.4.1. X-Ray Diffraction (XRD)

We used X-ray diffraction (XRD) in order to determine the dimensional parameters of crystals

2.4.2. Static contact angle

Surfaces can be characterized as being hydrophilic or hydrophobic depending on the value of the contact angle. An angle value less than 90° means that the surface has a hydrophilic behaviour; while an angle above 90° characterize a hydrophobic surface.

2.5. Biological test

We used the blood smear to examine the blood cells (leuckocytes, eritrocites and platelets) for any modification in size and shape according to known standards [28, 29]. We have chosen fresh blood from a healthy pacient on EDTA vacuum tubes, having normal hemoleucograme parameters that can be seen in table 1, obtained by couning with an automatic hematology analizer type Sismex.

Table 1. Hemoleucogramme parameters

<i>Parameter</i>	<i>Result</i>	<i>Reference values</i>
WBC (LEUCOCITE)	7.35 [10 ³ /uL]	4.00 - 11.00
RBC(ERITROCITE)	5.01 [10 ⁶ /uL]	4.00 - 6.30
HGB(HEMOGLOBINA)	13.8 [g/dL]	12.00 - 18.00
HCT(HEMATOCRIT)	41.2 [%]	37.00 - 51.00
MCV	82.2 [fL]	80.00 - 97.00
MCH	27.5 [pg]	26.00 - 32.00)
MCHC	33.5 [g/dL]	31.00 - 36.00
PLT (TROMBOCITE)	204 [10 ³ /uL]	140-440)
NEUT%	79.0 [%]	37.00 - 92.0
LYMPH%	15.1[%]	10.00 - 58.0
MONO%	4.4 [%]	0.0 - 14.0
EO%	1.2 [%]	0.0 - 6.0
BASO%	0.3 [%]	0.0 - 1.0

The materials were placed in Petri dishes in humid environment in order to avoid blood dehydration, then 200 microliters blood was added on the materials surface.

The Petri dishes were incubated for 30 minutes, respectively 60 minutes at 37°C. The smears were prepared using 5 microliters of blood from the fresh blood and at each interval from all samples. The smears were colored using May-Grumwald-Giemsa method using reagents from Merck, and the immersion reading was made with an optical microscope.

2.6. Electrodeposition of HA/ SWCNT-COOH/Fe hybrid material

The electro-deposition resulted curves are presented in Fig. 1. It can be seen that they are similar to the ones reported in the literature by having the signal divided into 3 time sections. As expected, we obtained the same results as previously mentioned in the literature [11]. Further the hybrid ceramic material structure was studied by X-Ray Diffraction.

3. Results and discussion

3.1. X-Ray Diffraction (XRD)

The samples were analysed with D8 DISCOVER, Bruker diffractometer equipped with Gobeil mirror, LynxEye 0D detector and Cu tube. We used the below measuring conditions: 0.040 increment, the scanning speed was 3s / pas and theta tangential incidence of 10. Measuring range was between 200 - 800.

Based on the qualitative analysis phase we identified the compounds with Ti, Fe, C, Ca, O, standing out C (Swcnts), TiO_2 , $\text{Fe}(\text{OH})_2$, CaFe_2O_4 , $\text{Fe}_2\text{Ti}_3\text{O}_9$ structures.

For both our samples, the C (Swcnts) compound have a hexagonal structure and this appears at $2\theta = 250, 430$ and 530 , being better shaped for $\text{TiO}_2\text{HA-Fe-Swcnt}$ 0.01M sample (Table 2, Fig. 2).

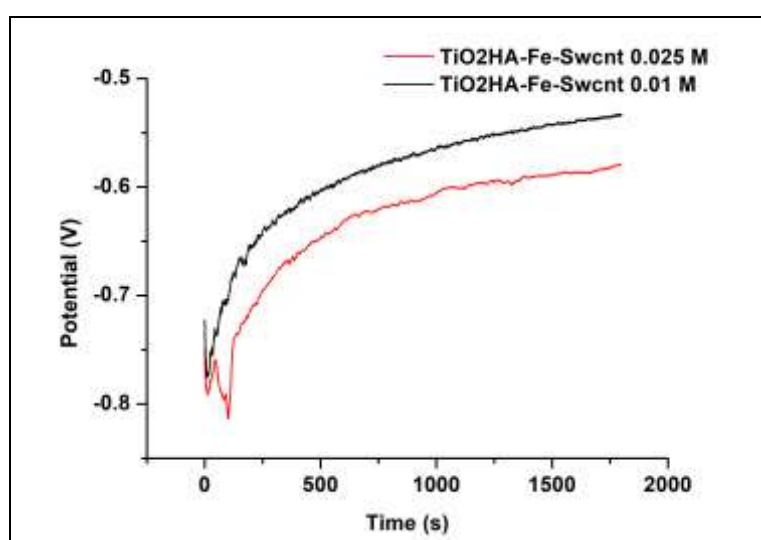


Fig. 1. The electrodeposition curve for HA onto TiO_2 on titanium plates

Table 2. Crystallite average size calculation and elementary cell parameters for $\text{TiO}_2\text{HA-Fe-Swcnt}$ 0.01M and $\text{TiO}_2\text{HA-Fe-Swcnt}$ 0.025M samples

Compounds	Crystallographic phase	Crystallite average size	Crystallization system	Elementary cell parameters
$\text{TiO}_2\text{HA-Fe-Swcnt}$ 0.01M	$\text{Ti}_{0.325}$	$D = 178.4 \text{ \AA}$	Hexagonal	$a = 2.9700;$ $c = 4.7751$
	$\text{Fe}_{0.96}(\text{OH})$	$D = 181.6 \text{ \AA}$	Hexagonal	$a = 2.9630 ;$ $c = 9.3700$
	C	$D = 461.8 \text{ \AA}$	Hexagonal	$a = 2.4285;$ $c = 6.9279$
$\text{TiO}_2\text{HA-Fe-Swcnt}$ 0.025M	Ti	$D = 206.8 \text{ \AA}$	Hexagonal	$a = 2.9700;$ $c = 4.7200$
	Fe_4N	$D = 353.9 \text{ \AA}$	Cubic	$a = b = c = 3.3800$
	$\text{Fe}_2\text{Ti}_3\text{O}_9$	$D = 340.3 \text{ \AA}$	Hexagonal	$a = 2.8667;$ $c = 4.5985$

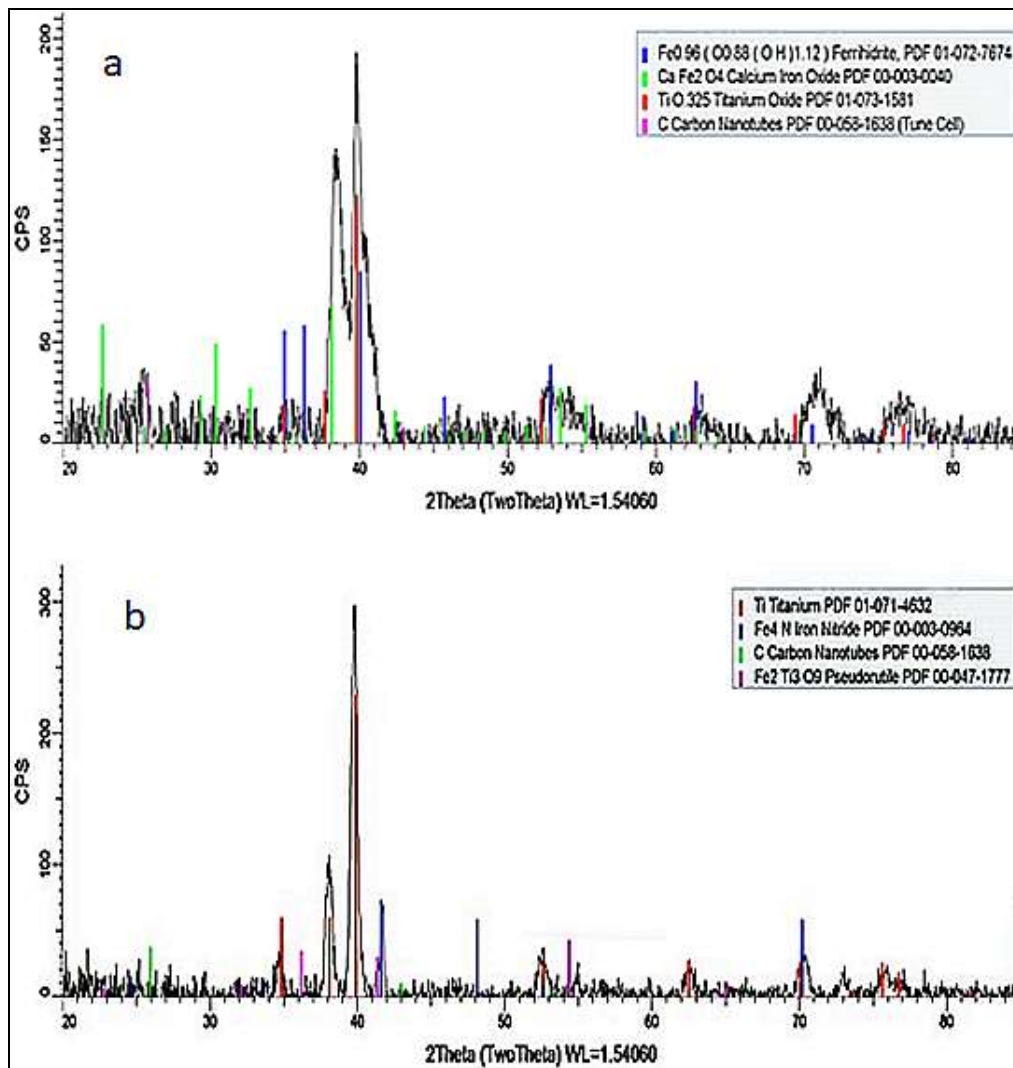


Fig. 2. XRD pattern for: a) TiO₂HA-Fe-Swcnt 0.01M and b) TiO₂HA-Fe-Swcnt 0.025 M

3.2. Static contact angle

We performed the wettability studies in order to characterise the nature of the surface. The contact angle measurements were performed at room temperature. The tests were performed with distilled water. The obtained results for both samples are represented in Table 3. In this case the recorded values show a strong hydrophilic character.

Table 3. Contact angle values for TiO₂HA-Fe-Swcnt 0.01 M and TiO₂HA-Fe-Swcnt 0.025 M samples

<i>Materials</i>	<i>Contact angle</i>	<i>STDEV %</i>
TiO ₂ HA-Fe-Swcnt 0.01 M	21.48	1.4
TiO ₂ HA-Fe-Swcnt 0.025 M	27,38	1.9

3.3. Electrochemical behaviour

The EIS measurement were performed at open circuit potential and were presented in Bode plots in Fig. 3a and Fig. 3b.

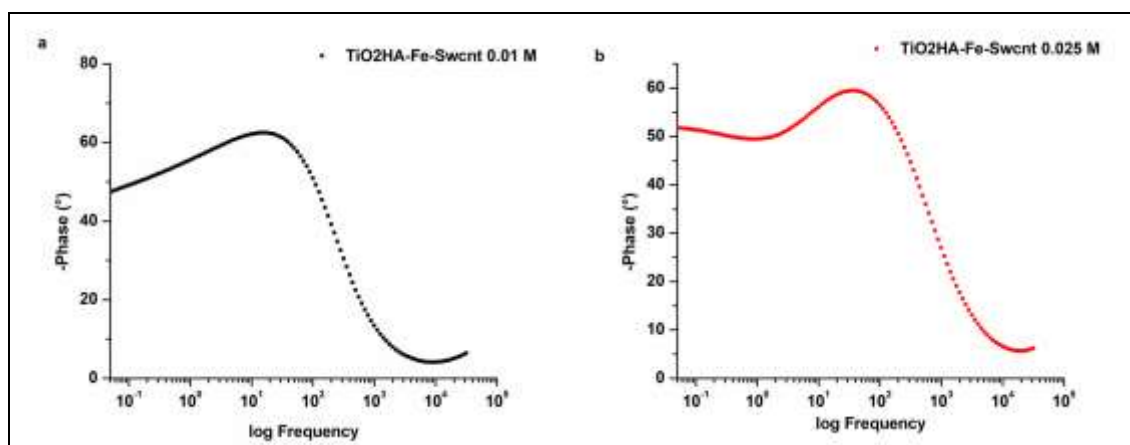


Fig. 3. Bode plots for: a) TiO₂HA-Fe-Swcnt 0.01 M sample and b) TiO₂HA-Fe-Swcnt 0.025 M sample in Hank solution

Both materials TiO₂HA-Fe-Swcnt 0.01 M and TiO₂HA-Fe-Swcnt 0.025 M have the phase angle at 50 degree for low frequencies and increase to medium frequencies at 60 degrees, having a diffusive behavior with capacitive tendency. It can be seen that the phase angle for high frequencies is decreasing. The materials have the same behavior as TiO₂HA-Fe-0.025 material.

Electrochemical study was followed by potentiodynamic measurements conducted in Hank solution presented in Fig. 4.

From Table 4 we can notice that the corrosion rates for both sample is in domain of perfect stable values. The EIS obtained dates were fitted with Nova 1.7 software, obtaining the circuit from Fig. 5.

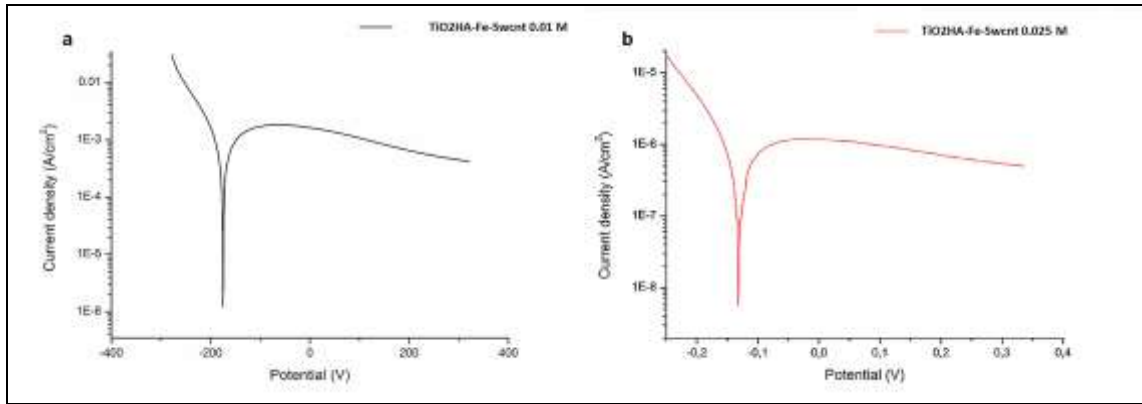


Fig. 4. Polarization curve for: a) TiO₂HA-Fe-Swcnt 0.01 M sample and b) TiO₂HA-Fe-Swcnt 0.025 M sample in Hank solution

Table 4. Corrosion parameters from Tafel polarization curves obtained for TiO₂HA-Fe-SWCNTs 0.01 and TiO₂HA-Fe-SWCNTs 0.025 samples in Hank solution

Sample	E_{cor} (V)	j_{corr} (A/cm ²)	i_{cor} (A)	Corrosion rate, (mm/year)	Polarization resistance (Ω)
TiO ₂ HA-Fe-Swcnt 0.01 M	- 0.17433	6.1162*10 ⁻⁷	6.9174E*10 ⁻⁷	0.0053225	29835
TiO ₂ HA-Fe-Swcnt 0.025 M	- 0.13162	6.0431*10 ⁻⁷	6.8348E*10 ⁻⁷	0.005259	39244

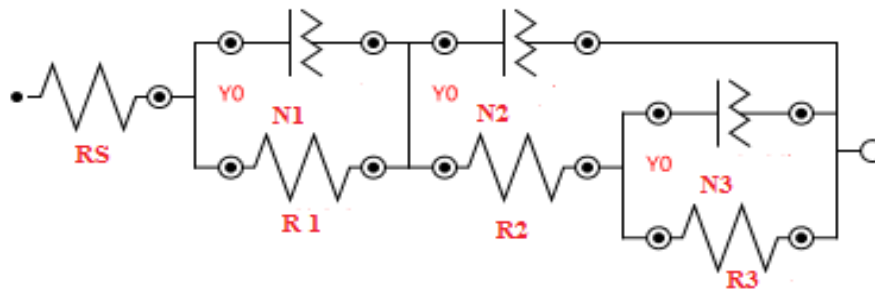


Fig. 5. The circuits for EIS data fit for TiO₂HA-Fe-Swcnt 0.01 M and TiO₂HA-Fe-Swcnt 0.025 M samples

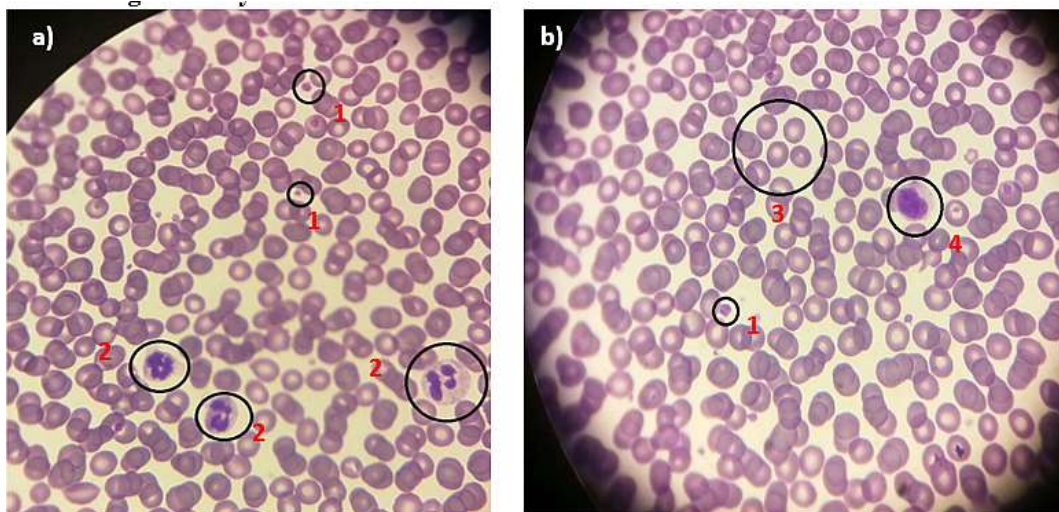
R_s is the solution resistance, R_1 the resistance of barrier layer, R_2 is the resistance of the porous layer, R_3 is the polarization resistance of the apatite layer. The obtained parameters are shown in Table 5.

Table 5. The fitting values obtained from the equivalent circuit for TiO₂HA-Fe-SWCNTs-0.01, TiO₂HA-Fe-SWCNTs-0.025 samples in Hank solution.

<i>Material</i>	R_s (Ω)	R_1 (Ω)	n_1	R_2 (Ω)	n_2	R_3 (Ω)	n_3
TiO ₂ HA-Fe-Swcnt 0.01 M	9.08	50.5	0.815	918	0.888	515×10^3	0.49
TiO ₂ HA-Fe-Swcnt 0.025 M	9.01	46.2	0.996	317×10	0.8	144×10^4	0.572

3.4. Biological analysis

First two images (Fig. 6) present the first smear used as control, in order to observe the structure and shape of the normal cells before having contact with the materials. In the fig. 6 a are highlighted three neutrophiles, two platelets and normal erythrocytes, in 6 b a monocyte, a platelet and some red blood cells. As can be seen, the membrane of the RBC are neat, smooth edges, and one third of the cells are empty. So, the blood chosen for testing is normal, the erythrocyte, leukocyte and platelet series are normal it is to mention that some authors have correlated the blood-contacting behaviour of surfaces with their interfacial free energy and surface chemistry [30].

**Fig. 6.** The control smear (1-platelets, 2-neutrophiles, 3-red blood cells, 4-monocyte)

For TiO₂HA-Fe 0.01 sample we can observe after 30 minutes (Fig. 7 a, b, c) that the lymphocyte and neutrophiles did not change the shape, but the trombocytes start to change, they developed pseudopodes, in order to be able to adhere to the material. After one hour (Fig. 7 d, e, f), also no changes in RBC, neutrophiles and lymphocytes, the trombocytes have pseudopodes.

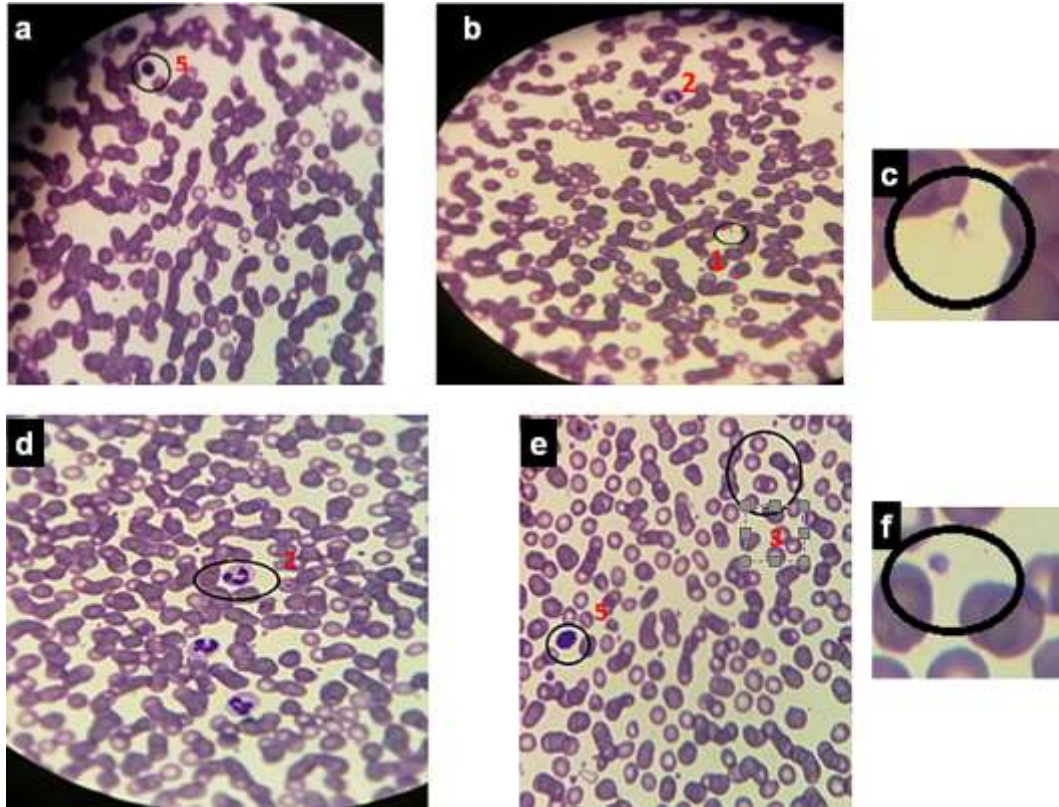


Fig. 7. TiO₂HA-Fe 0.01 sample

30 minutes with: a) normal lymphocyte (5), b) with normal neutrophils (2) and platelets with pseudopodes (1), c) platelet with pseudopodes; and 60 minutes with: d) normal neutrophils (2), e) normal lymphocyte (5), red blood cells (3) f) platelet with pseudopodes

For sample TiO₂HA-Fe 0.025 (Fig. 8) can be observed the same type of behavior like sample TiO₂HA-Fe 0.01 sample, for both 30 and 60 minutes, no modification in blood cells shape, except pseudopodes in thrombocytes.

For TiO₂HA-Fe-Swcnt 0.01 M sample (Fig. 9 a) and TiO₂HA-Fe-Swcnt 0.025 M sample (Fig. 10 a) in the first 30 minutes nothing happened, but after 60 minutes (Fig. 9 b and 10 b) we observe the apparition of echinocytes, red blood cells with scalloped edges, due to a slight degradation of RBC in contact with the material. Pseudopodes in thrombocytes are present, so they adhere to the material.

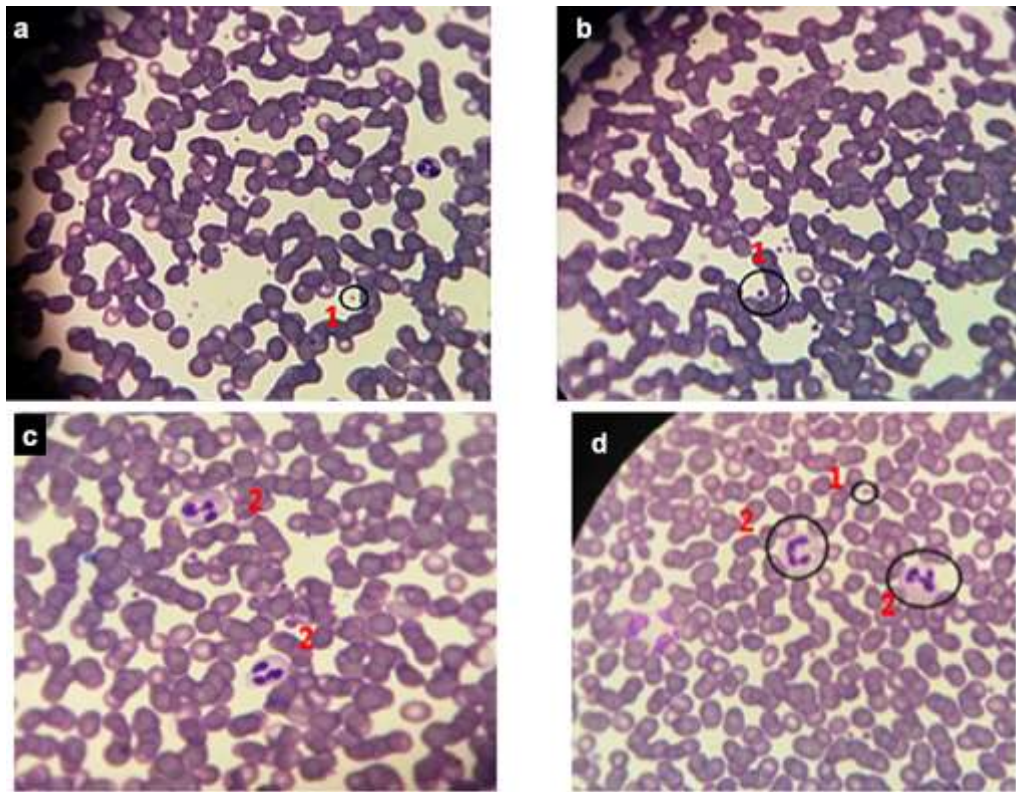


Fig. 8. TiO₂HA-Fe 0.025 sample
30 minutes with: a) neutrophils and platelets, b) platelets;
and 60 minutes with: c) normal neutrophils, d) thrombocytes having pseudopods

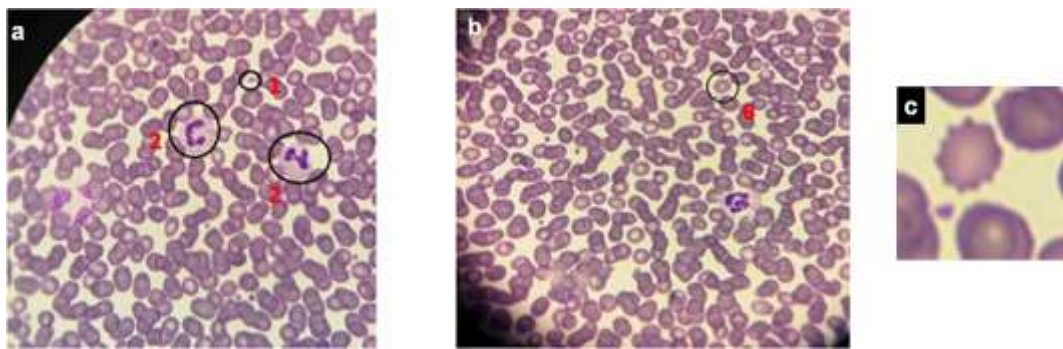


Fig. 9. TiO₂HA-Fe-Swnt 0.01 M: a) 30 minutes (normal elements), b) 60 minutes (6-echinocyte)
and c) echinocytes

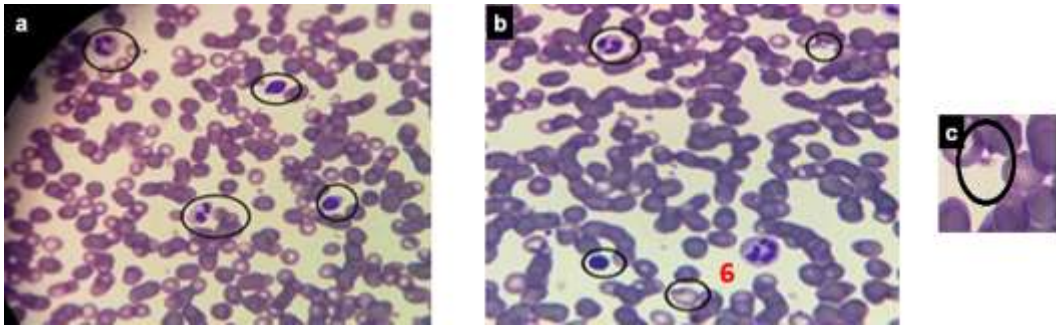


Fig. 10. TiO₂HA-Fe-Swcnt 0.025 M: a) 30minutes, b) 60minutes (6-echinocytes) and c) Trombocytes with pseudopodes

Conclusions

In this study we investigated the corrosion behaviour of a new composite TiO₂ HA-Fe-Swcnt 0.025 M material versus TiO₂HA-Fe-Swcnt 0.01 M.

The contact angles measurements for both investigated samples, reveals a strong hydrophilic behavior.

The samples structure was investigated via XRD analysis and we obtained a hexagonal structure for the C (SWCNTs) compound from both our samples. The corrosion rate values for all coatings immersed in Hank solution at 37 °C are in the perfect stable domain of corrosion resistance.

After biological tests we can notice no sign of evident toxic effect was observed, meaning no signs of hemolysis were observed in all samples, only a slight degradation of RBC after one hour of contact and also, no tendency to forming agregates in case of platelets was observed. The pseudopodes were produced in all samples very quickly, even after 30 minutes.

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